the plane of the three O atoms. Clearly, spatial accommodation of the sodium-halide pair is much more important than the precise direction of bonds within the $\mathrm{NaO}_{3} \mathrm{~A}$ groups. The Na atom has a marked anisotropy of thermal motion in the direction of the sodium-halide bond which may be related to the rather large thermal vibration of the halide ion. The oxygen thermal motion is markedly anisotropic normal to the plane of the $\mathrm{Al}-\mathrm{O}, \mathrm{Ge}-\mathrm{O}$ bond plane, as expected.

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# Structure of Tin Hafnium Sulfide and Lead Hafnium Sulfide 

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#### Abstract

SnHfS}_{3}, M_{r}=393.36\), orthorhombic, Pnma, $a=9.139$ (1), $b=3.694$ (1), $c=13.875$ (4) $\AA, \quad V=$ 468.4 (2) $\AA^{3}, Z=4, D_{x}=5.578 \mathrm{~g} \mathrm{~cm}{ }^{3}, \lambda($ Mo $K \bar{\alpha})=$ $0.71073 \AA, \mu=284.1 \mathrm{~cm}^{-1}, F(000)=680, T=298 \mathrm{~K}$, $R_{F}=0.036$ for 1301 observed reflections with $I \geq 2 \cdot 5 \sigma(I) . \quad \mathrm{PbHfS}_{3}, \quad M_{r}=481 \cdot 88$, orthorhombic, Pnma, $a=8.988$ (2), $b=3.739$ (1), $c=13.924$ (2) $\AA$, $V=467.9(2) \AA^{3}, \quad Z=4, D_{x}=6.840 \mathrm{gcm}{ }^{3}, \lambda($ Мо $K \bar{\imath})$ $=0.71073 \AA, \quad \mu=593.0 \mathrm{~cm}^{-1}, \quad F(000)=808, \quad T=$ $298 \mathrm{~K}, R_{F}=0.036$ for 1343 observed reflections with $I \geq 2.5 \sigma(I)$. The compounds are isostructural with $\mathrm{PbZrS}_{3}$ with modified $\mathrm{NH}_{4} \mathrm{CdCl}_{3}$ structure. There are columns running along the $b$ axis of double edgesharing octahedra $\left(\mathrm{Hf}_{2} \mathrm{~S}_{6}\right)_{\omega}$; these columns are linked by Sn or Pb with coordinations typical for $\mathrm{Sn}^{2+}$ and $\mathrm{Pb}^{2+}$ with lone-pair $s^{2}$ electrons.


Introduction. Structures of the ternary sulfides $A B S_{3}$ ( $A=\mathrm{Pb} ; B=\mathrm{Zr}, \mathrm{Hf}$ ) have been studied using powder methods by Sterzel \& Horn (1970), Yamaoka \& Okai

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(1970) and Lelieveld \& IJdo (1978). The latter authors concluded from a neutron powder diffraction refinement of $\mathrm{PbZrS}_{3}$ that the compound, and probably also $\mathrm{PbHfS}_{3}$, is isostructural with $\mathrm{PbSnS}_{3}$ and $\mathrm{Sn}_{2} \mathrm{~S}_{3}$. The structures are strongly related to the $\mathrm{NH}_{4} \mathrm{CdCl}_{3}$ structure (Strukturbericht, type E 24). In a study of the ternary sulfides with the assumed composition $A B S_{3}$ ( $A=\mathrm{Sn}, \mathrm{Pb} ; B=\mathrm{Nb}, \mathrm{Ta}$ ) we found that single crystals could be obtained by vapour transport using chlorine as transport agent. We found this method also suitable for compounds $\mathrm{PbHfS}_{3}$ and $\mathrm{SnHfS}_{3}$, the latter being a new compound. Structure determinations using single crystals are reported.

Experimental. Powder samples of $\mathrm{SnHfS}_{3}$ and $\mathrm{PbHfS}_{3}$ were obtained by heating the elements in evacuated quartz ampoules. An intimate mixture of the elements was first slowly heated to about 673 K for two days during which the binary sulfides are probably formed. The ampoules were then heated at 1023 K for ten days. In this way the reaction of the metals with quartz from the wall was avoided. $\mathrm{PbHfS}_{3}$ was then heated for three

[^1]days at 1123 K . X-ray powder data of the samples showed that the compounds were probably isostructural. Single crystals were grown by vapour transport in a temperature gradient of 1043 to 983 K . To about 300 mg of the compounds about 10 mg $\left(\mathrm{NH}_{4}\right)_{2} \mathrm{PbCl}_{6}$ was added. This compound decomposes at high temperature into $\mathrm{PbCl}_{2}, \mathrm{NH}_{4} \mathrm{Cl}$ and $\mathrm{Cl}_{2} . \mathrm{Cl}_{2}$ probably acts as transport agent. The crystals grow as bundles of red needles at the low-temperature side of the quartz ampoules; the needle axes correspond to the $b$ axes of the orthorhombic unit cells.

Suitable red needle-shaped crystals were glued on the top of glass fibres and mounted on an Enraf-Nonius CAD-4F diffractometer, with graphite monochromator. Unit-cell dimensions and their standard deviations were determined from the setting angles of 18 reflections in the range $27<\theta<30^{\circ}$ in four alternate settings (de Boer \& Duisenberg, 1984). All reflections were measured in one quadrant of the Ewald sphere up to $\theta=40^{\circ}$. Three check reflections were measured every 2 h of X-ray exposure time; these exhibit no significant decay during data collection. A $360^{\circ} \psi$ scan for the close to axial reflection (040) showed an intensity variation up to $30 \%$ for $\mathrm{SnHfS}_{3}$ and up to $32 \%$ for $\mathrm{PbHfS}_{3}$ about the mean value. The intensity data were corrected for the scale variation, Lorentz and polarization effects and for absorption using a Gaussian integration method (Spek, 1983). Standard deviations in the intensities based on counting statistics were increased according to an analysis of the excess variance of the three reference reflections: $\sigma^{2}(I)=\sigma_{\mathrm{cs}}^{2}(I)+(P I)^{2}$ where $P$ is the instability constant (McCandlish, Stout \& Andrews, 1975). The space group was derived from the observed systematic absences.

Initial atomic parameters were taken from Lelieveld \& IJdo (1978). Refinement on $F$ by full-matrix least-squares techniques with anisotropic thermal parameters converged at $R_{F}=0.036(w R=0.028)$ for $\mathrm{SnHfS}_{3}$ and $R_{F}=0.037(w R=0.045)$ for $\mathrm{PbHfS}_{3}$, respectively. The crystals exhibited some secondary extinction for which the $F$ values were corrected by refinement of an empirical isotropic extinction parameter (Zachariasen, 1967). The final difference Fourier map did not show any significant residual features. Crystal data and experimental details of the structure determination are compiled in Table 1.* Final fractional atomic coordinates and equivalent isotropic thermal parameters for the atoms are given in Table 2. Bond distances and angles are listed in Table 3. Scattering factors were taken from Cromer \& Mann

[^2]Table 1. Details of the structure determination

|  | $\mathrm{SnHfS}_{3}$ | $\mathrm{PbHfS}_{3}$ |
| :---: | :---: | :---: |
| Data collection |  |  |
| Crystal dimensions (mm) | $0.007 \times 0.042 \times 0.42$ | $0.02 \times 0.03 \times 0.60$ |
| $\theta$ range; min.. max. ${ }^{\circ}{ }^{\circ}$ ) | 1.47, 40.0 | 1.46, 40.0 |
| (1)/2A scan ( ${ }^{\circ}$ ) | $\Delta \omega=0.90+0.35 \tan \theta$ | $1.20+0.35 \tan \theta$ |
| Data set | $\begin{gathered} h: 0 \rightarrow 16 ; k:-0 \rightarrow 6 ; \\ \quad:-25 \rightarrow 25 \end{gathered}$ | $h:-16 \rightarrow 16 ; k:-0 \rightarrow 6:$ |
| Crystal-to-receiving-aperture distance (mm) | 173 | 173 |
| Horizontal, vertical aperture (mm) | 4.0.4.5 | 4.0.4.5 |
| Reference reflections, r.m.s. | 202, 0.63 | 0.66 |
| deviation (\%) | 202, 0.63 | 0.69 |
|  | 211,0.66 | 0.80 |
| Instability constant $P$ | 0.005 | 0.015 |
| Drift correction | 1.00-1.00 | 0.992-1.000 |
| Min. and max. absorption correction factor | 1.23-3.07 | 2.18-5.32 |
| X-ray exposure time (h) | 66.5 | 83.1 |
| Total data | 3365 | 6446 |
| Unique data | 1610 | 1611 |
| Observed data $\|I \geq 2 \cdot 5 \sigma(I)\|$ | 1301 | 1343 |
| $R_{1} I=\stackrel{ \pm}{\Sigma}(l-\bar{l}) / \pm I \mid$ | 0.034 | 0.111 |
| $R_{2}[=\Sigma \sigma / \Sigma I]$ | 0.054 | 0.056 |
| Number of equivalent reflections | 3266 | 6440 |
| Refinement |  |  |
| Number of reflections | 1301 | 1343 |
| Number of refined parameters | 32 | 32 |
| Isotropic secondary-extinction coefficient, $g$ | $0.0075(6) \times 10^{4}$ | $0.0137(8) \times 10^{4}$ |
| Final agreement factors |  |  |
| $R_{F}$ | 0.036 | 0.037 |
| $w R$ | 0.028 | 0.045 |
| Weighting scheme | $1 / \sigma^{\prime}(F)$ | 1 |
| $S$ | 2.058 | 3.930 |
| Residual electron density in final difference Fourier map (e $\AA^{3}$ ) | -4.5,6.1 | $-4.8 .5 \cdot 8$ |
| Max. (shift/ $\sigma$ ) final cycle | $2.3 \times 10^{3}$ | $6.2 \times 10^{\circ}$ |
| Average (shift/ $\sigma$ ) final cycle | $1.7 \times 10^{4}$ | $8.2 \times 10^{7}$ |

Table 2. Final fractional atomic coordinates and equivalent isotropic thermal parameters with e.s.d.'s in parentheses

(1968). Anomalous-dispersion factors were taken from Cromer \& Liberman (1970). All calculations carried out on the CDC-Cyber 170/760 computer of the University of Groningen with the program packages XTAL (Hall \& Stewart, 1987), and EUCLID (Spek, 1982; calculation of geometric data).

Discussion. The structures of $\mathrm{SnHfS}_{3}$ and $\mathrm{PbHfS}_{3}$ reported here, $\mathrm{PbZrS}_{3}$ (Lelieveld \& IJdo, 1978), $\mathrm{PbSnS}_{3}$ (Jumas, Ribes, Philippot \& Maurin, 1972), $\mathrm{Sn}_{2} \mathrm{~S}_{3}$ (Mootz \& Puhl, 1967) are strongly related to the $\mathrm{NH}_{4} \mathrm{CdCl}_{3}$ ( E 24 ) structure with columns of double edge-sharing octahedra, $\left(\mathrm{Cd}_{2} \mathrm{Cl}_{6}\right)_{\infty}$, running along the $b$

Table 3. Selected data on the geometry; distances in $\AA$, angles in ${ }^{\circ}$

Standard deviations in the last decimal place are given in parentheses. Bonds and angles occurring in pairs are indicated by *.

|  | $\mathrm{SnHfS}_{3}$ | $\mathrm{PbHfS}_{3}$ |
| :--- | :---: | :---: |
| $\mathrm{Hf}-\mathrm{S}(1)$ | $2.501(2)$ | $2.488(3)$ |
| $\mathrm{Hf}-\mathrm{S}(2)^{*}$ | $2.530(2)$ | $2.533(2)$ |
| $\mathrm{Hf}-\mathrm{S}(3)^{*}$ | $2.577(2)$ | $2.582(2)$ |
| $\mathrm{Hf}-\mathrm{S}(3)$ | $2.567(2)$ | $2.569(3)$ |
| $M-\mathrm{S}(1)^{*}$ | $2.613(2)$ | $2.729(3)$ |
| $M-\mathrm{S}(2)$ | $2.758(2)$ | $2.907(3)$ |
| $M-\mathrm{S}(3)$ | $3.082(2)$ | $3.073(3)$ |
| $M-\mathrm{S}(2)^{*}$ | $3.490(2)$ | $3.383(3)$ |
| $M-\mathrm{S}(1)^{*}$ | $3.722(2)$ | $3.546(3)$ |
|  |  |  |
| $\mathrm{S}(1)-M-\mathrm{S}(1)$ | $89.97(5)$ | $86.48(7)$ |
| $\mathrm{S}(1)-M-\mathrm{S}(2)^{*}$ | $85.18(6)$ | $81.68(7)$ |
| $\mathrm{S}(1)-M-\mathrm{S}(3)^{*}$ | $79.08(6)$ | $78.10(7)$ |

Note: only angles between the shortest bonds are given.


Fig. 1. Projection of the structure of $\mathrm{SnHfS}_{3}$ and $\mathrm{PbHfS}_{3}$ along [010]. Open and hatched circles are atoms at $\frac{1}{4}$ and $\frac{3}{4}$, respectively; bonds between atoms indicated by open and hatched circles occur in pairs.
axis. In $\mathrm{SnHfS}_{3}$ and $\mathrm{PbHfS}_{3}$ the columns of double edge-sharing octahedra $\left(\mathrm{Hf}_{2} \mathrm{~S}_{6}\right)_{\infty}$ are linked by Sn and Pb respectively (Fig. 1). The $\mathrm{Hf}-\mathrm{S}$ distances are in the range 2.501 (2) to 2.577 (2) $\AA$ for $\mathrm{SnHfS}_{3}$ and 2.488 (3) to 2.582 (2) $\AA$ for $\mathrm{PbHfS}_{3}$. Sn and Pb atoms have three short bonds to $S(1)$ and $S(2): 2 \cdot 613(2)(2 \times)$ and 2.758 (2) $\AA$, respectively, for $\mathrm{SnHfS}_{3}, 2.729$ (3) ( $2 \times$ ) and 2.907 (3) $\AA$, respectively, for $\mathrm{PbHfS}_{3}$. The bond angles, $\mathrm{S}-\mathrm{M}-\mathrm{S}$, are close to $90^{\circ}$. If the somewhat longer $\mathrm{Sn}-\mathrm{S}(3)[3.082$ (2) $\AA$ ] distance and the $\mathrm{Pb}-\mathrm{S}(3)$ $[3.073$ (3) $\AA]$ distance are taken into account, the coordination is distorted square pyramidal. With $S$ atoms at still larger distances, the coordination is a distorted dicapped trigonal prism with Sn and Pb in a very asymmetric position. This asymmetric coordination with three strong bonds with angles close to $90^{\circ}$ is typical for $\mathrm{Sn}^{2+}$ and $\mathrm{Pb}^{2+}$ ions; the asymmetry is due to the presence of a lone pair of $s$ electrons ( $5 s$ for $\mathrm{Sn}^{2+}, 6 s$ for $\left.\mathrm{Pb}^{2+}\right)$. It is seen that the differences between the relatively small $M-\mathrm{S}$ distances $(M=\mathrm{Sn}, \mathrm{Pb})$ and
the large $M-S$ distances are larger for $\mathrm{SnHfS}_{3}$ than for $\mathrm{PbHfS}_{3}$. This tendency is also visible in the structures of the low- and high-temperature forms of SnS (von Schnering \& Wiedemeier, 1981; Wiedemeier \& von Schnering, 1978) compared with PbS . The $\mathrm{S}(1), \mathrm{S}(2)$ and $S(3)$ atoms each have three close neighbours. $S(1)$ has one close Hf and two close Pb neighbours; $\mathrm{S}(2)$ has two close Hf neighbours and one close Pb neighbour while $\mathrm{S}(3)$ has three close Hf neighbours. The coordination of the atoms in $\mathrm{SnHfS}_{3}$ and $\mathrm{PbHfS}_{3}$ is in agreement with ions $\mathrm{Sn}^{2+}\left(\mathrm{Pb}^{2+}\right)$ and $\mathrm{Hf}^{4+}$. Preliminary conductivity measurements showed the compounds to be semiconductors. Completely different structures are found for ternary sulfides ' $\mathrm{Sn} M \mathrm{~S}_{3}$ ' and ${ }^{\prime} \mathrm{PbMS}_{3}$ ' ( $M$ $=\mathrm{Nb}, \mathrm{Ta}$ ). Recent structure determinations (Meetsma, Wiegers, Haange \& de Boer, 1989; Wiegers, Meetsma, Haange \& de Boer, 1988) have shown that these compounds are misfit layer compounds consisting of two-atom-thick slabs of PbS and SnS with Pb and Sn in distorted square-pyramidal coordination and sandwiches $\mathrm{NbS}_{2}$ and $\mathrm{TaS}_{2}$ as also occur in $\mathrm{NbS}_{2}-2 H$ and $\mathrm{TaS}_{2}-2 H$ ( Nb and Ta in trigonal-prismatic coordination with sulfur). The composition of these compounds, $(\mathrm{SnS})_{n} M \mathrm{~S}_{2}, \quad(\mathrm{PbS})_{n} M \mathrm{~S}_{2}, \quad n=1 \cdot 13-1 \cdot 17$, is determined by the misfit between the $\mathrm{SnS}(\mathrm{PbS})$ and the $\mathrm{NbS}_{2}\left(\mathrm{TaS}_{2}\right)$ parts of the structure.

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[^2]:    * Lists of structure factors, anisotropic thermal parameters, bond lengths and angles have been deposited with the British Library Document Supply Centre as Supplementary Publication No. SUP 51683 (19 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

