the plane of the three O atoms. Clearly, spatial accommodation of the sodium-halide pair is much more important than the precise direction of bonds within the NaO₃A groups. The Na atom has a marked anisotropy of thermal motion in the direction of the sodium-halide bond which may be related to the rather large thermal vibration of the halide ion. The oxygen thermal motion is markedly anisotropic normal to the plane of the Al–O,Ge–O bond plane, as expected.

This study was supported by a Natural Sciences and Engineering Research Council of Canada operating grant.

References

- BARBIER, J. & FLEET, M. E. (1987). J. Solid State Chem. 71, 361-370.
- BARRER, R. M. & VAUGHAN, D. E. W. (1971). J. Phys. Chem. Solids, 32, 731-743.

BEAGLEY, B., HENDERSON, C. M. B. & TAYLOR, D. (1982). Mineral. Mag. 46, 459-464.

COPPENS, P. & HAMILTON, W. C. (1970). Acta Cryst. A 26, 71-83.

- HASSAN, I. & GRUNDY, H. D. (1984). Acta Cryst. B40, 6-13.
- International Tables for X-ray Crystallography (1974). Vol. IV. Birmingham: Kynoch Press. (Present distributor Kluwer Academic Publishers, Dordrecht.)
- KLASKA, K.-H. (1974). Strukturuntersuchungen an Tridymitabkommlingen (Beryllonitreihe, Trinephelin), p. 103. Dissertation, Univ. of Hamburg, Federal Republic of Germany.
- LIEBAU, F. (1985). In *Structural Chemistry of Silicates*. Berlin: Springer-Verlag.
- McLaughlan, S. D. & Marshall, D. J. (1970). Phys. Lett. A, 32, 343-344.
- MEGAW, H. D. (1973). Crystal Structures. Philadelphia: W. B. Saunders.
- MEIER, W. M. (1968). Zeolite Structures. In Molecular Sieves, pp. 10–27. London: Society of Chemistry for Industry.
- MERLINO, S. (1984). Feldspathoids: Their average and real structures. In Feldspars and Feldspathoids, NATO ASI Ser. C, Vol. 137, pp. 435–470. Dordrecht: Reidel.
- NYMAN, H. & HYDE, B. G. (1981). Acta Cryst. A 37, 11-17.
- PAULING, L. (1930). Z. Kristallogr. 74, 213-225.
- TAYLOR, D. (1972). Mineral. Mag. 38, 593-604.

Acta Cryst. (1989). C45, 847-849

Structure of Tin Hafnium Sulfide and Lead Hafnium Sulfide

By G. A. WIEGERS,* A. MEETSMA, R. J. HAANGE AND J. L. DE BOER

Laboratory of Inorganic Chemistry, Materials Science Centre of the University of Groningen, Nijenborgh 16. 9747 AG Groningen, The Netherlands

(Received 1 November 1988; accepted 15 December 1988)

Abstract. SnHfS₃, $M_r = 393 \cdot 36$, orthorhombic, *Pnma*, a = 9.139(1), b = 3.694(1), c = 13.875(4) Å, V =468.4 (2) Å³, Z = 4, $D_x = 5.578$ g cm⁻³, λ (Mo $K\bar{\alpha}$) = 0.71073 Å, $\mu = 284.1$ cm⁻¹, F(000) = 680, T = 298 K, $R_F = 0.036$ for 1301 observed reflections with $I \ge 2.5\sigma(I)$. PbHfS₃, $M_r = 481.88$, orthorhombic, *Pnma*, a = 8.988 (2), b = 3.739 (1), c = 13.924 (2) Å, $V = 467.9 (2) \text{Å}^3$, Z = 4, $D_x = 6.840 \text{ g cm}^{-3}$, $\lambda (\text{Mo} K \bar{a})$ = 0.71073 Å, $\mu = 593.0$ cm⁻¹, F(000) = 808, T =298 K, $R_F = 0.036$ for 1343 observed reflections with $I \ge 2 \cdot 5\sigma(I)$. The compounds are isostructural with PbZrS₃ with modified NH₄CdCl₃ structure. There are columns running along the b axis of double edgesharing octahedra $(Hf_2S_6)_{(i)}$; these columns are linked by Sn or Pb with coordinations typical for Sn²⁺ and Pb^{2+} with lone-pair s^2 electrons.

Introduction. Structures of the ternary sulfides ABS_3 (A = Pb; B = Zr, Hf) have been studied using powder methods by Sterzel & Horn (1970), Yamaoka & Okai

0108-2701/89/060847-03\$03.00

(1970) and Lelieveld & IJdo (1978). The latter authors concluded from a neutron powder diffraction refinement of PbZrS₃ that the compound, and probably also PbHfS₃, is isostructural with PbSnS₃ and Sn₂S₃. The structures are strongly related to the NH₄CdCl₃ structure (*Strukturbericht*, type E 24). In a study of the ternary sulfides with the assumed composition ABS_3 (A = Sn, Pb; B = Nb, Ta) we found that single crystals could be obtained by vapour transport using chlorine as transport agent. We found this method also suitable for compounds PbHfS₃ and SnHfS₃, the latter being a new compound. Structure determinations using single crystals are reported.

Experimental. Powder samples of SnHfS₃ and PbHfS₃ were obtained by heating the elements in evacuated quartz ampoules. An intimate mixture of the elements was first slowly heated to about 673 K for two days during which the binary sulfides are probably formed. The ampoules were then heated at 1023 K for ten days. In this way the reaction of the metals with quartz from the wall was avoided. PbHfS₃ was then heated for three

© 1989 International Union of Crystallography

^{*} Author to who correspondence should be addressed.

days at 1123 K. X-ray powder data of the samples showed that the compounds were probably isostructural. Single crystals were grown by vapour transport in a temperature gradient of 1043 to 983 K. To about 300 mg of the compounds about 10 mg $(NH_4)_2PbCl_6$ was added. This compound decomposes at high temperature into PbCl₂, NH₄Cl and Cl₂. Cl₂ probably acts as transport agent. The crystals grow as bundles of red needles at the low-temperature side of the quartz ampoules; the needle axes correspond to the *b* axes of the orthorhombic unit cells.

Suitable red needle-shaped crystals were glued on the top of glass fibres and mounted on an Enraf-Nonius CAD-4F diffractometer, with graphite monochromator. Unit-cell dimensions and their standard deviations were determined from the setting angles of 18 reflections in the range $27 < \theta < 30^{\circ}$ in four alternate settings (de Boer & Duisenberg, 1984). All reflections were measured in one quadrant of the Ewald sphere up to $\theta = 40^{\circ}$. Three check reflections were measured every 2 h of X-ray exposure time; these exhibit no significant decay during data collection. A $360^{\circ} \psi$ scan for the close to axial reflection (040) showed an intensity variation up to 30% for SnHfS₃ and up to 32% for PbHfS, about the mean value. The intensity data were corrected for the scale variation, Lorentz and polarization effects and for absorption using a Gaussian integration method (Spek, 1983). Standard deviations in the intensities based on counting statistics were increased according to an analysis of the excess variance of the three reference reflections: $\sigma^2(I) = \sigma_{cs}^2(I) + (PI)^2$ where P is the instability constant (McCandlish, Stout & Andrews, 1975). The space group was derived from the observed systematic absences.

Initial atomic parameters were taken from Lelieveld & IJdo (1978). Refinement on F by full-matrix least-squares techniques with anisotropic thermal parameters converged at $R_F = 0.036$ (wR = 0.028) for SnHfS₃ and $R_F = 0.037$ (wR = 0.045) for PbHfS₃, respectively. The crystals exhibited some secondary extinction for which the F values were corrected by refinement of an empirical isotropic extinction parameter (Zachariasen, 1967). The final difference Fourier map did not show any significant residual features. Crystal data and experimental details of the structure determination are compiled in Table 1.* Final fractional atomic coordinates and equivalent isotropic thermal parameters for the atoms are given in Table 2. Bond distances and angles are listed in Table 3. Scattering factors were taken from Cromer & Mann

Table 1. Details of the structure determination

	SnHfS,	PbHfS,
Data collection		
Crystal dimensions (mm)	$0.007 \times 0.042 \times 0.42$	$0.02 \times 0.03 \times 0.60$
θ range; min., max. (°)	1.47.40.0	1.46.40.0
$\omega/2\theta$ scan (°)	$\Delta \omega = 0.90 + 0.35 \tan \theta$	$1.20 + 0.35 \tan \theta$
Data set	$h: 0 \rightarrow 16; k: -0 \rightarrow 6;$	$h: -16 \rightarrow 16; k: -0 \rightarrow 6;$
	1: -25→25	1:-25→25
Crystal-to-receiving-aperture distance (mm)	173	173
Horizontal, vertical aperture (mm)	4.0.4.5	4.0.4.5
Reference reflections, r.m.s.	202.0.63	0.66
deviation (%)	202.0.63	0-69
. ,	211, 0.66	0.80
Instability constant P	0.005	0-015
Drift correction	1.00-1.00	0.992-1.000
Min. and max. absorption correction factor	1.23-3.07	2.18-5.32
X-ray exposure time (h)	66.5	83-1
Total data	3365	6446
Unique data	1610	1611
Observed data $ I \ge 2 \cdot 5\sigma(I) $	1301	1343
$R_{1} = \sum (I - \overline{I}) / \sum I$	0.034	0.111
$R_2[=\Sigma\sigma/\Sigma I]$	0-054	0.056
Number of equivalent reflections	3266	6440
Refinement		
Number of reflections	1301	1343
Number of refined parameters	32	32
lsotropic secondary-extinction coefficient, g	0-0075 (6) × 10 4	0.0137 (8) × 10 4
Final agreement factors		
R _F	0.036	0.037
wR	0.028	0.045
Weighting scheme	$1/\sigma^2(F)$	1
S	2.058	3-930
Residual electron density in final difference Fourier map (e Å ⁻³)	-4-5, 6-1	-4-8.5-8
Max. (shift/ σ) final cycle	2.3×10^{-3}	6·2 × 10 *
Average (shift/ σ) final cycle	1.7 × 10 ⁻⁴	8·2 × 10 '

Table 2. Final fractional atomic coordinates andequivalent isotropic thermal parameters with e.s.d.'s inparentheses

x	у	Z	$U_{eq}^{*}(\dot{A}^{2})$
0.53676 (7)	1	0.82850 (5)	0.0161 (2)
0.16246 (4)	ł	0.05048 (3)	0.0095 (1)
0.2724 (2)	Ĩ	0.2155 (2)	0.0125 (5)
0.1697 (2)	Ĩ	0-4930 (2)	0-0098 (4)
0.0146 (2)	i	0.8931 (2)	0.0085 (5)
0-51515 (6)	1	0.82589 (4)	0.0175 (1)
0.16611 (6)	į	0.05057 (4)	0.0100 (1)
0.2731 (4)	į.	0.2154 (2)	0.0130 (6)
0.1683 (3)	į.	0.4902 (2)	0.0104 (6)
0.0130 (3)	14	0.8948 (2)	0.0092 (6)
	x 0.53676 (7) 0.16246 (4) 0.2724 (2) 0.1697 (2) 0.0146 (2) 0.51515 (6) 0.16611 (6) 0.2731 (4) 0.1683 (3) 0.0130 (3)	$\begin{array}{c ccccccccccccccccccccccccccccccccccc$	$\begin{array}{c ccccccccccccccccccccccccccccccccccc$

* $U_{eq} = \frac{1}{3} \sum_{i} \sum_{j} U_{ij} a_i a_j \mathbf{a}_i \cdot \mathbf{a}_j$

(1968). Anomalous-dispersion factors were taken from Cromer & Liberman (1970). All calculations carried out on the CDC-Cyber 170/760 computer of the University of Groningen with the program packages *XTAL* (Hall & Stewart, 1987), and *EUCLID* (Spek, 1982; calculation of geometric data).

Discussion. The structures of SnHfS₃ and PbHfS₃ reported here, PbZrS₃ (Lelieveld & IJdo, 1978), PbSnS₃ (Jumas, Ribes, Philippot & Maurin, 1972), Sn₂S₃ (Mootz & Puhl, 1967) are strongly related to the NH₄CdCl₃ (E 24) structure with columns of double edge-sharing octahedra, $(Cd_2Cl_6)_{\infty}$, running along the *b*

^{*} Lists of structure factors, anisotropic thermal parameters, bond lengths and angles have been deposited with the British Library Document Supply Centre as Supplementary Publication No. SUP 51683 (19 pp.). Copies may be obtained through The Executive Secretary, International Union of Crystallography, 5 Abbey Square, Chester CH1 2HU, England.

Table 3. Selected data on the geometry; distances in Å, angles in °

Standard deviations in the last decimal place are given in parentheses. Bonds and angles occurring in pairs are indicated by *.

	SnHfS ₃	PbHfS,
Hf-S(1)	2.501 (2)	2.488 (3)
Hf-S(2)*	2.530 (2)	2.533 (2)
Hf-S(3)*	2.577 (2)	2.582 (2)
Hf-S(3)	2.567 (2)	2.569 (3)
M-S(1)*	2.613 (2)	2.729 (3)
M-S(2)	2.758 (2)	2.907 (3)
M-S(3)	3.082 (2)	3.073 (3)
M-S(2)*	3.490 (2)	3.383 (3)
M-S(1)*	3.722 (2)	3.546 (3)
S(1) - M - S(1)	89.97 (5)	86-48 (7)
$S(1) - M - S(2)^*$	85-18 (6)	81.68 (7)
$S(1) - M - S(3)^*$	79.08 (6)	78.10(7)

Note: only angles between the shortest bonds are given.



Fig. 1. Projection of the structure of SnHfS₃ and PbHfS₃ along [010]. Open and hatched circles are atoms at $\frac{1}{4}$ and $\frac{3}{4}$, respectively; bonds between atoms indicated by open and hatched circles occur in pairs.

axis. In SnHfS₃ and PbHfS₃ the columns of double edge-sharing octahedra $(Hf_2S_6)_{\infty}$ are linked by Sn and Pb respectively (Fig. 1). The Hf-S distances are in the range 2.501(2) to 2.577(2)Å for SnHfS₃ and 2.488 (3) to 2.582 (2) Å for PbHfS₃. Sn and Pb atoms have three short bonds to S(1) and S(2): 2.613 (2) (2×) and 2.758 (2) Å, respectively, for SnHfS₃, 2.729 (3) $(2\times)$ and 2.907 (3) Å, respectively, for PbHfS₃. The bond angles, S-M-S, are close to 90°. If the somewhat longer Sn-S(3) [3.082 (2) Å] distance and the Pb-S(3) [3.073 (3) Å] distance are taken into account, the coordination is distorted square pyramidal. With S atoms at still larger distances, the coordination is a distorted dicapped trigonal prism with Sn and Pb in a very asymmetric position. This asymmetric coordination with three strong bonds with angles close to 90° is typical for Sn^{2+} and Pb^{2+} ions; the asymmetry is due to the presence of a lone pair of s electrons (5s for Sn^{2+} , 6s for Pb²⁺). It is seen that the differences between the relatively small M-S distances (M = Sn, Pb) and the large M-S distances are larger for SnHfS, than for PbHfS₂. This tendency is also visible in the structures of the low- and high-temperature forms of SnS (von Schnering & Wiedemeier, 1981; Wiedemeier & von Schnering, 1978) compared with PbS. The S(1), S(2) and S(3) atoms each have three close neighbours. S(1)has one close Hf and two close Pb neighbours; S(2) has two close Hf neighbours and one close Pb neighbour while S(3) has three close Hf neighbours. The coordination of the atoms in SnHfS, and PbHfS, is in agreement with ions Sn²⁺ (Pb²⁺) and Hf⁴⁺. Preliminary conductivity measurements showed the compounds to be semiconductors. Completely different structures are found for ternary sulfides 'SnMS₃' and 'PbMS₃' (M = Nb, Ta). Recent structure determinations (Meetsma, Wiegers, Haange & de Boer, 1989; Wiegers, Meetsma, Haange & de Boer, 1988) have shown that these compounds are misfit layer compounds consisting of two-atom-thick slabs of PbS and SnS with Pb and Sn in distorted square-pyramidal coordination and sandwiches NbS₂ and TaS₂ as also occur in NbS₂-2H and $TaS_{2}-2H$ (Nb and Ta in trigonal-prismatic coordination with sulfur). The composition of these compounds, $(SnS)_n MS_2$, $(PbS)_n MS_2$, $n = 1 \cdot 13 - 1 \cdot 17$, is determined by the misfit between the SnS (PbS) and the NbS_2 (TaS₂) parts of the structure.

References

- BOER, J. L. DE & DUISENBERG, A. J. M. (1984). Acta Cryst. A40, C410.
- CROMER, D. T. & LIBERMAN, D. (1970). J. Chem. Phys. 53, 1891-1898.
- CROMER, D. T. & MANN, J. B. (1968). Acta Cryst. A24, 321-324.
- HALL, S. R. & STEWART, J. M. (1987). Editors. XTAL2.2 User's Manual. Univs. of Western Australia, Australia, and Maryland, USA.
- JUMAS, J. C., RIBES, M., PHILIPPOT, E. & MAURIN, M. (1972). C. R. Acad. Sci. 275, 269–272.
- LELIEVELD, R. & IJDO, D. J. W. (1978). Acta Cryst. B34, 3348-3349.
- McCANDLISH, L. E., STOUT, G. H. & ANDREWS, L. C. (1975). Acta Cryst. A31, 245-249.
- MEETSMA, A., WIEGERS, G. A., HAANGE, R. J. & DE BOER, J. L. (1989). Acta Cryst. A45, 285-291.
- MOOTZ, D. & PUHL, H. (1967). Acta Cryst. 23, 471-476.
- SCHNERING, H. G. VON & WIEDEMEIER, H. (1981). Z. Kristallogr. 143-150.
- SPEK, A. L. (1982). The EUCLID package. In Computational Crystallography, edited by D. SAYRE, p. 528. Oxford: Clarendon Press.
- SPEK, A. L. (1983). Proc. 8th Eur. Crystallogr. Meet., Liège, Belgium.
- STERZEL, W. & HORN, J. (1970). Z. Anorg. Allg. Chem. 376, 254-260.
- WIEDEMEIER, H. & VON SCHNERING, H. G. (1978). Z. Kristallogr. 148, 295-303.
- WIEGERS, G. A., MEETSMA, A., HAANGE, R. J. & DE BOER, J. L. (1988). Int. Conf. Reactivity of Solids, Princeton. To appear in *Solid State Ionics*.
- YAMAOKA, S. & OKAI, B. (1970). Mater. Res. Bull. 5, 789-794.
- ZACHARIASEN, W. H. (1967). Acta Cryst. 23, 558-564.